

# Lecture 8

## Polymers and Gels

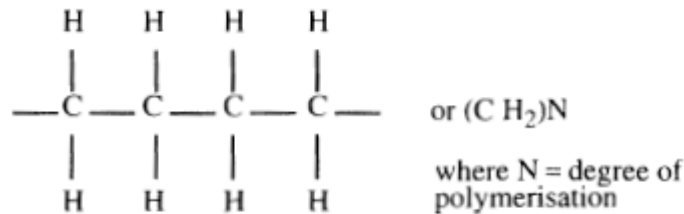
# Variety of polymeric materials

- Polymer – molecule made by repeating of covalently joint units.
- “Living polymers” (not considered in this lecture) – long-chain objects connected by physical forces e.g. giant worm-like micelles, actin filaments etc.
- Many of physical properties of polymers have **universal** characteristic related to generic properties of long string-like molecules

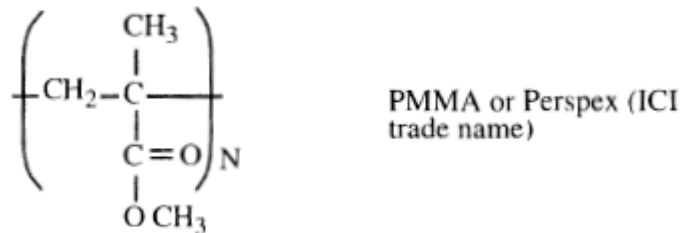
# Variety of polymeric materials

- Polymers are predominantly based on carbons though often contain other elements in the chain and sometimes don't involve carbon at all.

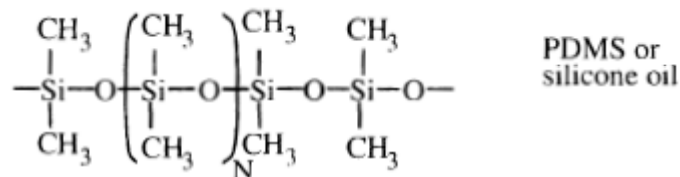
## Polyethylene



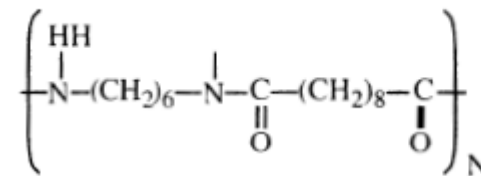
Polymethyl methacrylate



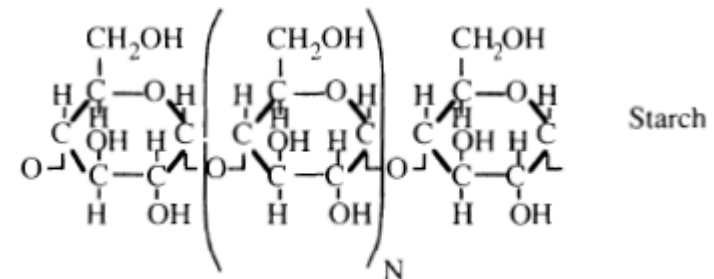
Polydimethyl siloxane



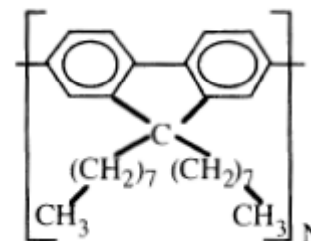
### Nylon 6-10



## Amylose

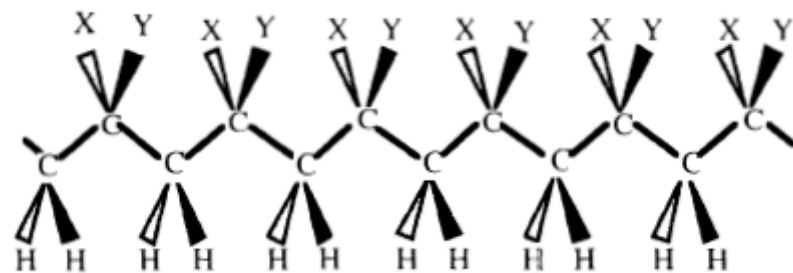


Poly(di-octyl fluorene)

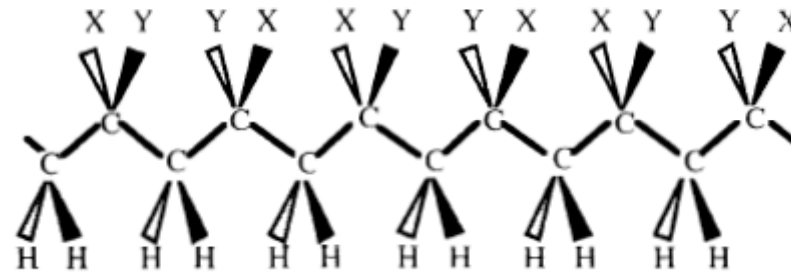


# Stereochemistry of polymers

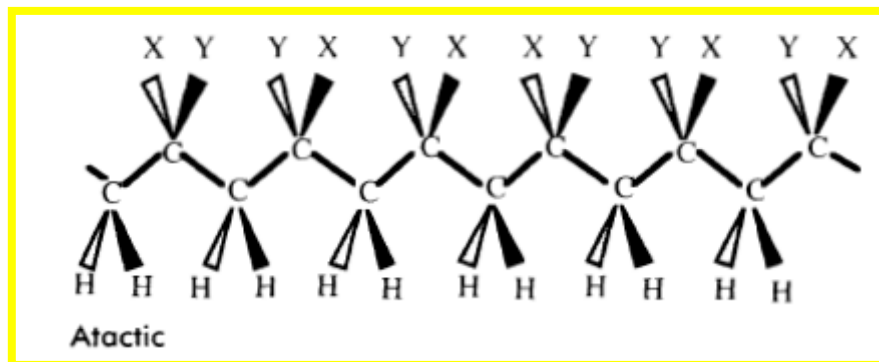
- If polymer has more than one type of side groups different arrangement are possible



Isotactic



Syndiotactic



Atactic

- Atactic** arrangement involves **quenched disorder** and therefore prevents crystallization: polymer will form glass at low temperature

# Architecture of polymers

- **linear or branched** ← **quenched disorder**
- average number of units in the chain:  
**degree of polymerization  $N$  or molecular mass  $M$**
- distribution of lengths: **polydispersity**
- formed of the same units:  
**homopolymers** or several types of units:  
**copolymers** ← **quenched disorder**
- copolymers: **random, sequenced**  
copolymers (DNA, proteins), **block**  
**copolymers** (units arranged in blocks) ← **microphase separation with complex morphologies**

**strictly prescribed  
sequence leading to  
self-assembly**

# Block copolymers in nanotechnology

- polymers where repeat units come in blocks:

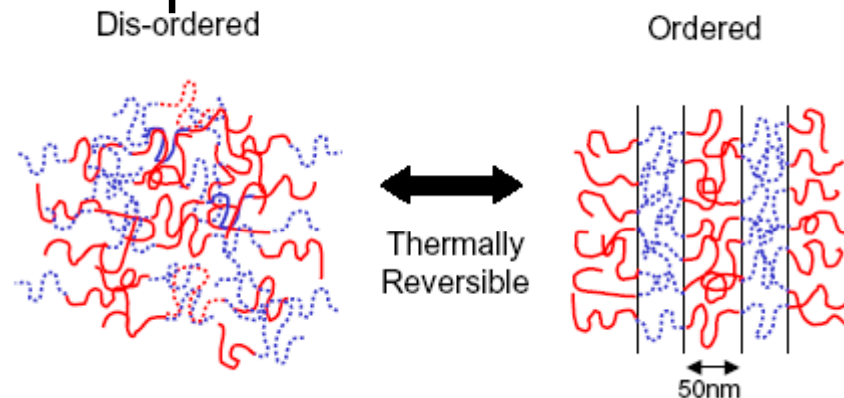
- AB di-block copolymers



- ABA tri-block copolymers

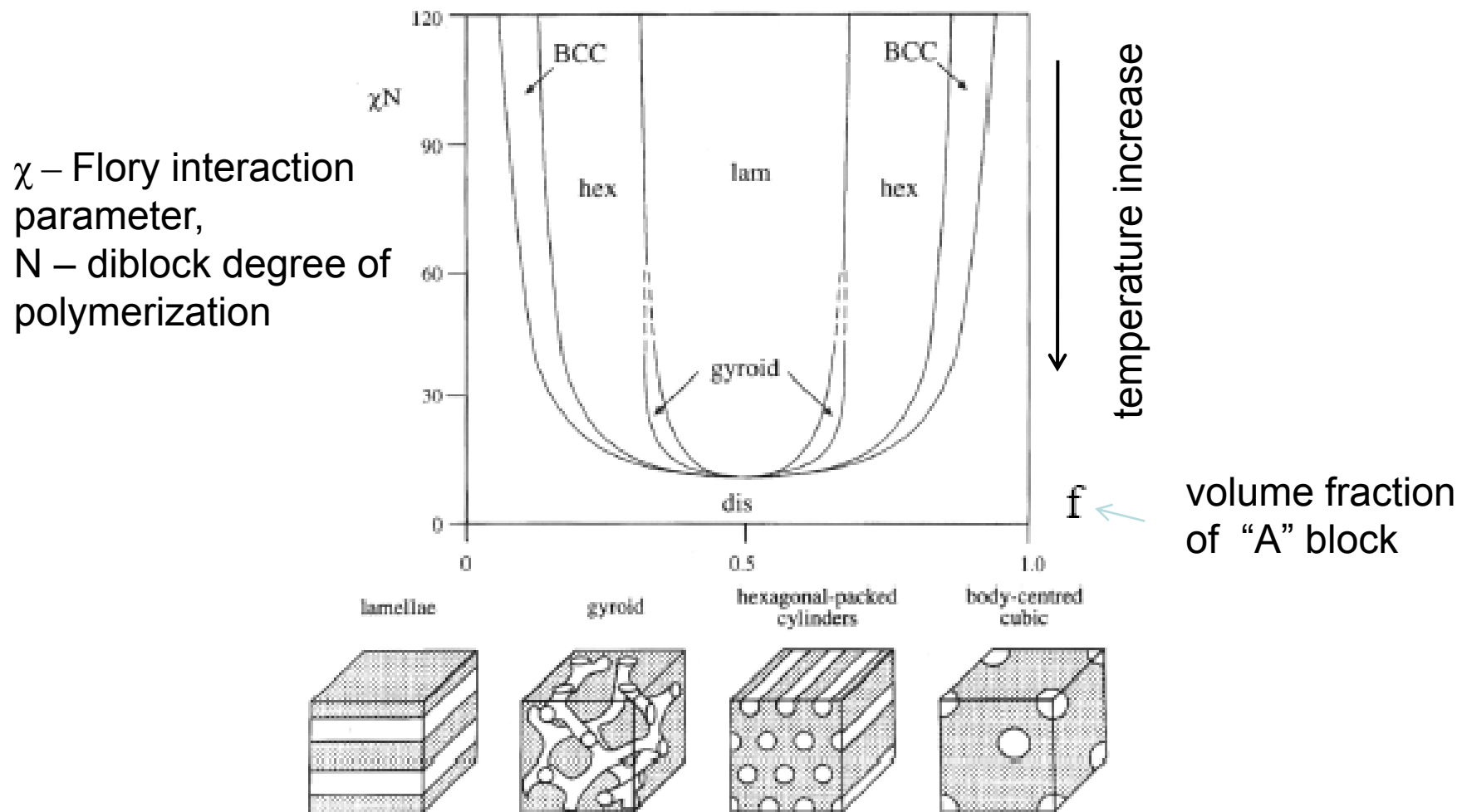


- can be combination of different polymers, e.g. amorphous (rubbery) and crystalline (hard)
- If units are very different chemically they will phase separate. This microphase separation leads to a zoo of possible structures



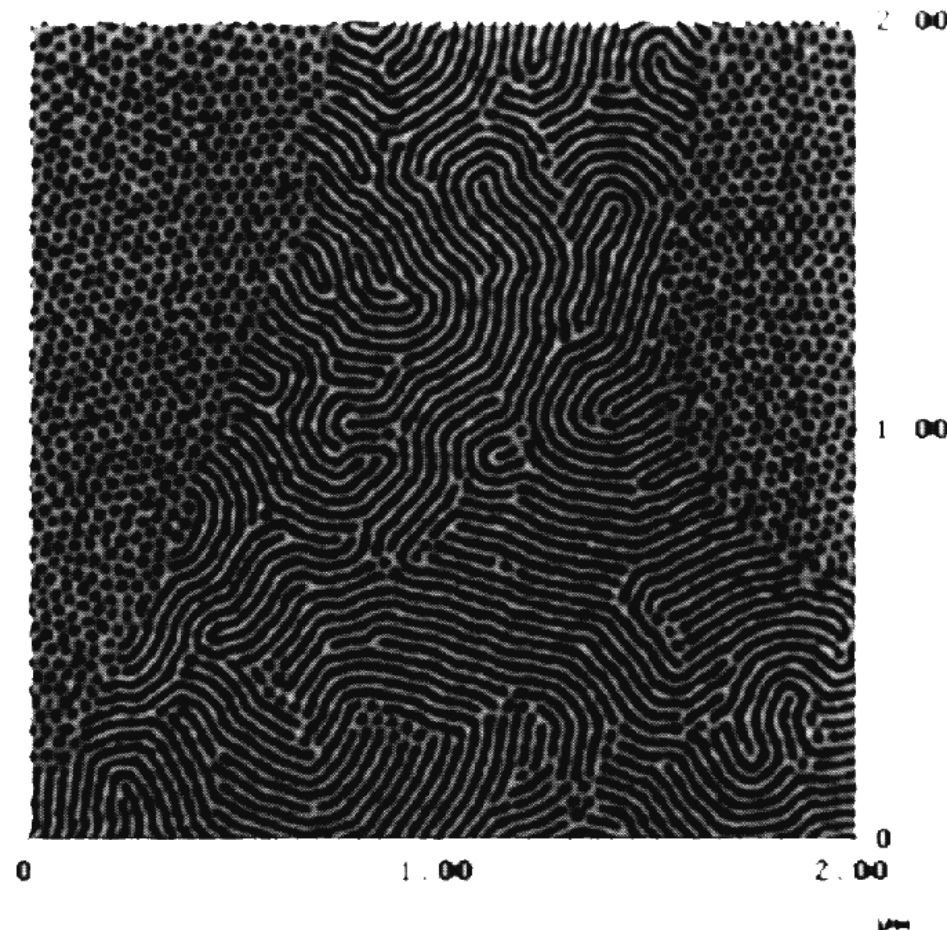
# Phase diagram of a block copolymer

- The structures formed depend on the relative length of blocks and compatibility of polymers



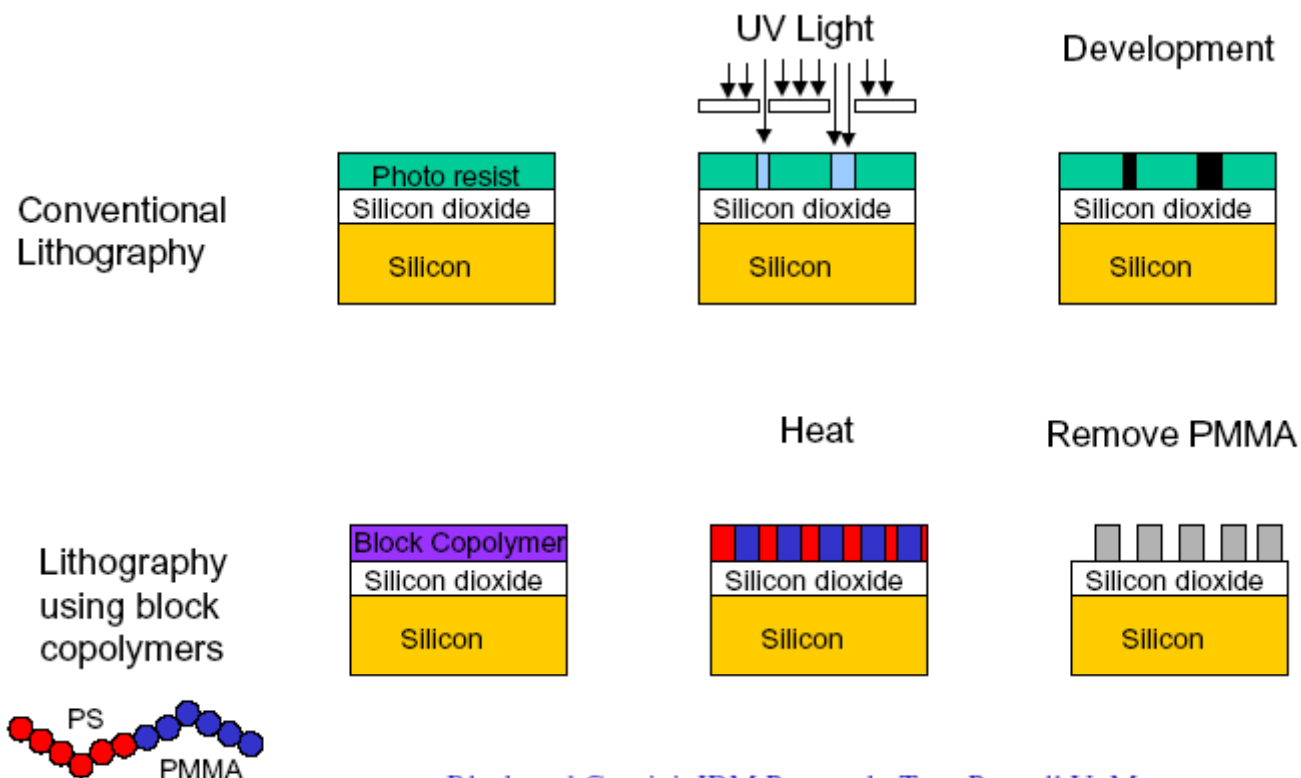
# Structure of block copolymers

- AFM of a triblock copolymer Kraton D-1102 (van Dijk and van der Berg, 1996)



# Nanopatterning with block copolymers

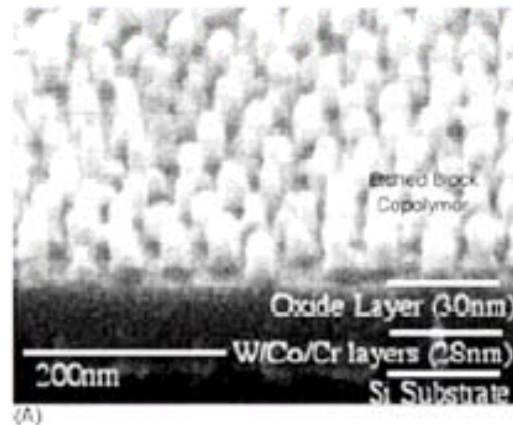
- provides cheap way to produce nanostructures, e.g. for magnetic storage



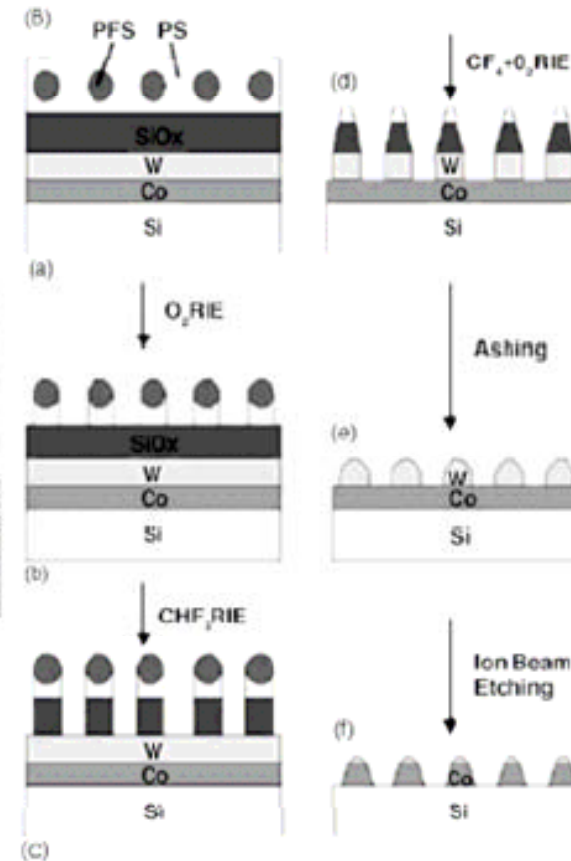
Black and Guarini, IBM Research. Tom Russell U. Mass.

# Creating nanodots

poly(styrene-b-ferrocenylsilane)  
block copolymer



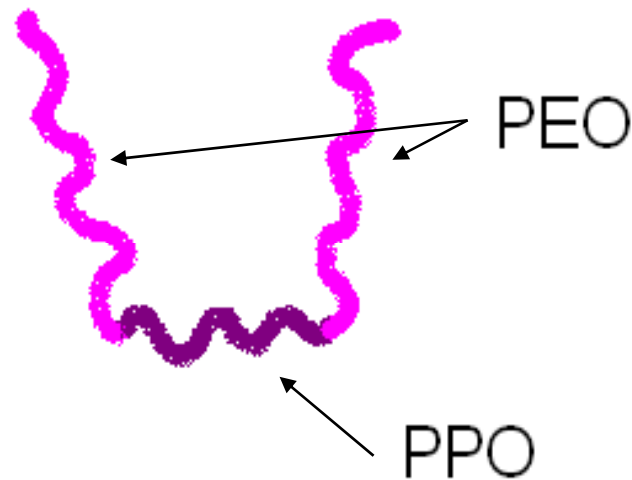
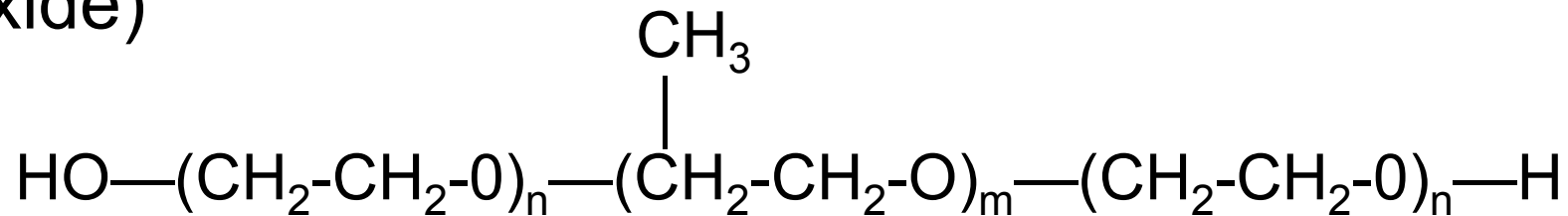
Structure applicable  
to hard drive  
manufacture



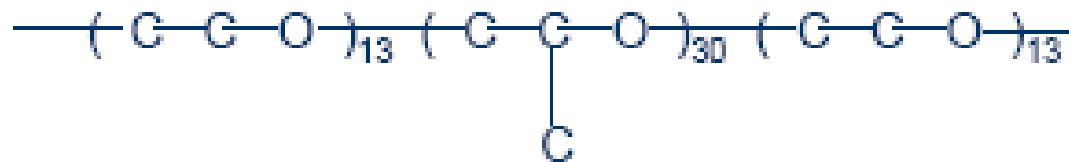
J.Y. Cheng, C.A. Ross, V.Z.H. Chan, E.L. Thomas, R.G.H. Lammertink, G.J. Vancso, Adv. Mater. 13, 1174–1178 (2001)

# Pluronics™ copolymers for drug delivery

- Symmetrical hydrophobically associating triblock copolymers, Poly(propylene oxide) and poly(ethylene oxide)



Pluronic L64™

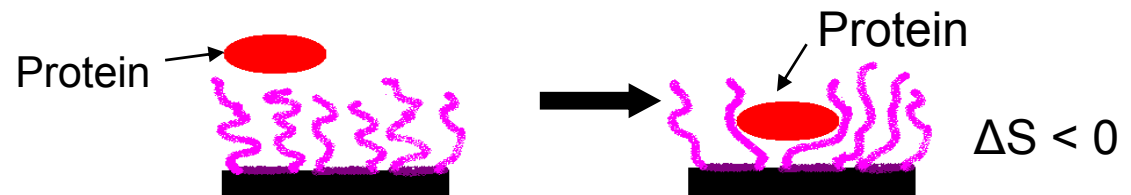


# Pluronics copolymers

- Poly(propylen oxide)
  - Central hydrophobic core
  - Folds in aqueous solution
  - CH<sub>3</sub> groups interact by Van der Waals
  - Binds hydrophobic proteins

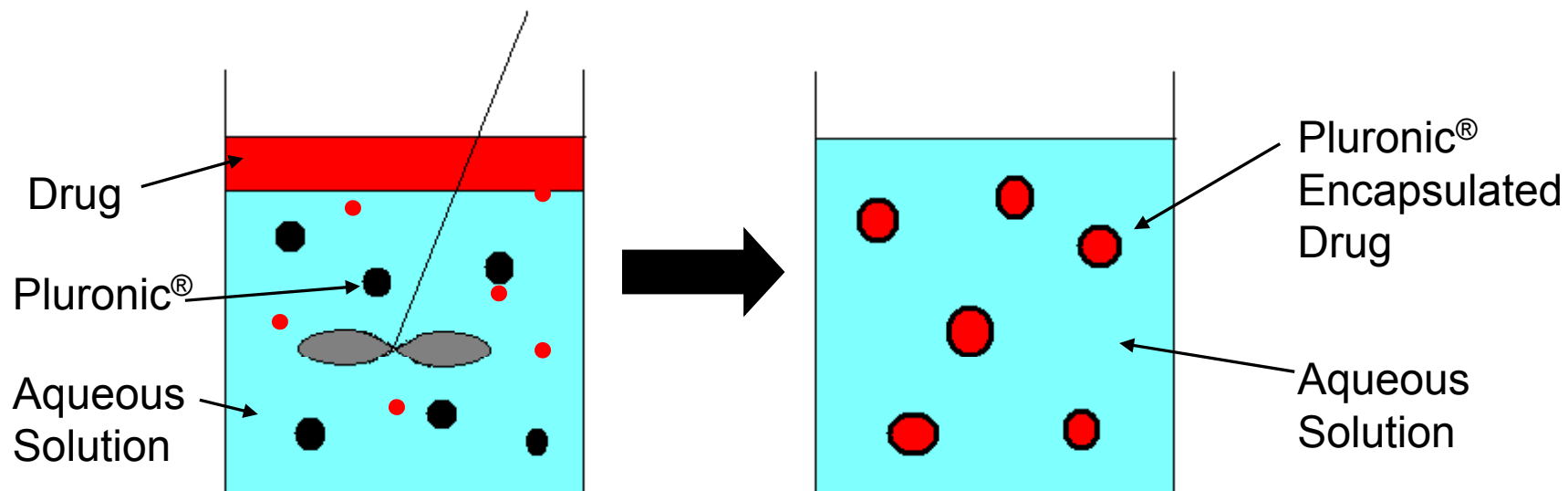
- Poly(ethylen oxide)

- Hydrophilic
- Soluble in water
  - Hydrogen bonding interaction
  - More PEO in Pluronic<sup>®</sup>, easier to dissolve
- Moves freely in aqueous solution
- High entropy → low protein adsorption



# Drug encapsulation by Pluronics

- Form after passing critical micelle concentration (CMC) or critical micelle temperature (CMT)
- Suspensions can encapsulate drugs

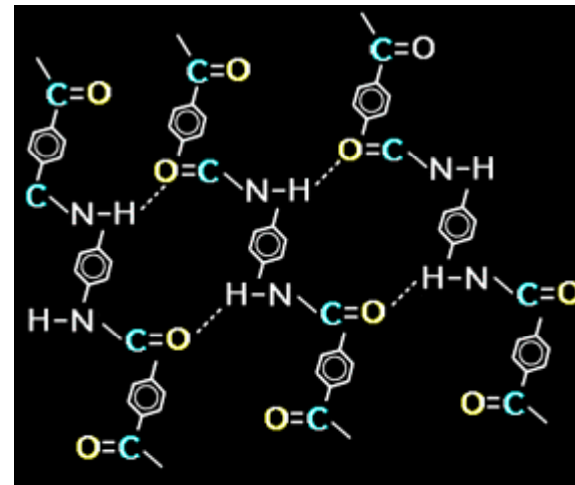
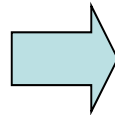
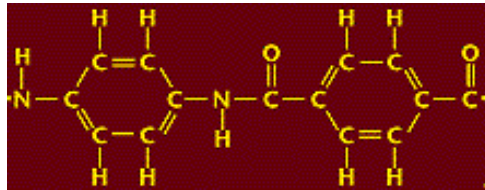


# Physical state of polymers

- **Liquid:** often very viscous with particular viscoelastic properties
- **Glass:** very common due to quenched disorder
- **Crystalline:** often semi-crystalline: micro crystallites in amorphous matrix
- **Liquid crystalline:** can be formed in a case of rigid molecules, e.g. Kevlar

# Kevlar

- Kevlar – aromatic polyamide
- chemically inert, mechanically strong, no degradation at low temperatures (down to  $-196^{\circ}\text{C}$ )



# Random walk and polymer chain

- **freely joint chain** polymer model – equivalent of random walk

- **end-to-end distance**  $\vec{r} = \sum_{i=1}^N \vec{a}_i$

- **mean square end-to-end distance**

$$\langle \vec{r} \cdot \vec{r} \rangle = \left\langle \left( \sum_{i=1}^N \vec{a}_i \right) \cdot \left( \sum_{j=1}^N \vec{a}_j \right) \right\rangle = \left\langle \sum_{i=1}^N \sum_{j=1}^N \vec{a}_i \cdot \vec{a}_j \right\rangle$$

$$\langle \vec{r}^2 \rangle = Na^2 + \left\langle \sum_{i \neq j}^N \vec{a}_i \cdot \vec{a}_j \right\rangle$$

0, as different links are **uncorrelated**

# Entropy of a polymer chain

- Entropy of a macrostate is related to the number of available microstates (statistical weight):

$$S = k_B \ln \Omega$$

# Entropy of a polymer chain

- In a case of 1D walk:  $R_x = (N_+ - N_-)a_x$ ;  $\Omega_x = \frac{N!}{N_+!N_-!}$

$$\ln x! = x \ln x - x; \quad \ln \Omega_x = -N [f \ln f + (1-f) \ln(1-f)], \quad f = N_+/N$$

- function has a maximum at  $f=1/2$ , let's make Taylor expansion around the maximum:

$$\ln \Omega_x = \ln \Omega_x(1/2) + (1/2 - f) \left( \frac{d \ln \Omega_x}{df} \right)_{f=1/2} + \frac{1}{2} (1/2 - f)^2 \left( \frac{d^2 \ln \Omega_x}{df^2} \right)_{f=1/2} + \dots$$

$$\ln \Omega_x \approx N \ln 2 - 2N (1/2 - f)^2; \quad \ln \Omega_x \approx -\frac{R_x^2}{2Na_x^2}$$

- in 3D:

$$\ln \Omega = \ln \Omega_x \Omega_y \Omega_z \approx -\frac{3R^2}{2Na^2}$$

# Entropy of a polymer chain

- Configurational entropy of a freely joint polymer chain:

$$S(r) \approx -\frac{3k_B R_x^2}{2Na_x^2} + \text{const}$$

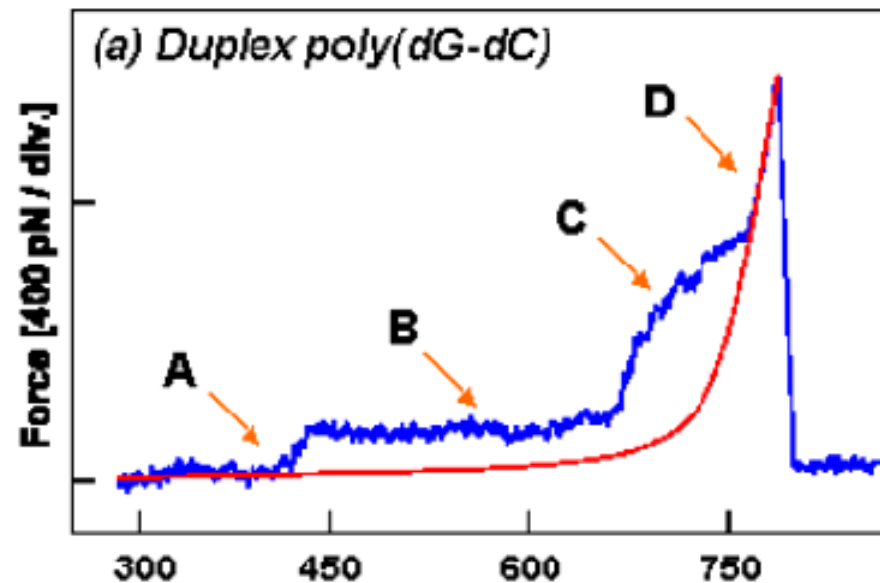
- Gibbs energy

$$\Delta G = -T\Delta S = \frac{3k_B T R_x^2}{2Na_x^2}$$

$$F = \frac{3k_B T}{Na_x^2} r$$

Entropic force, quite small

# Stretching a DNA



- Dario Anselmetti et al, "Single Molecule DNA Biophysics with Atomic Force Microscopy", SingleMolecules, 2001

## More realistic model: short range correlations

- realistically, correlation will not disappear for nearest neighbors due to e.g. bond angle

$$\left\langle \vec{a}_i \cdot \vec{a}_{i-1} \right\rangle = a^2 \cos \theta; \left\langle \vec{a}_i \cdot \vec{a}_{i-m} \right\rangle = a^2 (\cos \theta)^m$$

- the chain will behave as its subunits are effectively longer

$$\left\langle \vec{r}^2 \right\rangle = \frac{N}{g} \left\langle c^2 \right\rangle = N b^2$$

- correlations can be characterized by characteristic ratio

$$C_{\infty} = \frac{b^2}{a^2}$$

# Excluded volume: Flory's approach

- Consider a molecule that occupies volume  $r^3$ :

- segment concentration:  $c \approx \frac{N}{r^3}$

- This leads to reduction of entropy:  $\Delta G = \frac{k_B v N}{V} T = \frac{k_B v N^2}{2r^3} T$

- Total energy including configurational energy:

$$\Delta G = \Delta G_{rep} + \Delta G_{el} = \frac{k_B T r^2}{N a^2} + \frac{k_B T v N^2}{2 r^3}$$

- assuming excluded volume  $v \sim a^3$   
and minimizing energy:

$$r \sim a N^{3/5}$$

# Interaction between polymer segments

- Number of contacts for a polymer molecule:

$$N_{pp} = \frac{1}{2} zNvc; \quad N_{ps} = zNv(1-c); \quad N_{ss} = N_{ss}^0 - \frac{1}{2} zNvc - zNv(1-c)$$

$z$  – number of neighbors for any polymer segment or solvent molecule

- The interaction energy:

$$U_{\text{int}} = \frac{1}{2} zNvc(\varepsilon_{pp} + \varepsilon_{ss} - 2\varepsilon_{ps}) + zN(\varepsilon_{ps} + \varepsilon_{ss}) + N_{ss}^0 \varepsilon_{ss}$$

$$U_{\text{int}} = k_B T \chi 2v \frac{N^2}{2r^3} + \text{const}; \quad k_B T \chi = \frac{1}{2} z(\varepsilon_{pp} + \varepsilon_{ss} - 2\varepsilon_{ps})$$

- The total energy:

$$G_{\text{rep}} + U_{\text{int}} = k_B T v (1 - 2\chi) \frac{N^2}{2r^3} + \text{const};$$

# Interaction between polymer segments

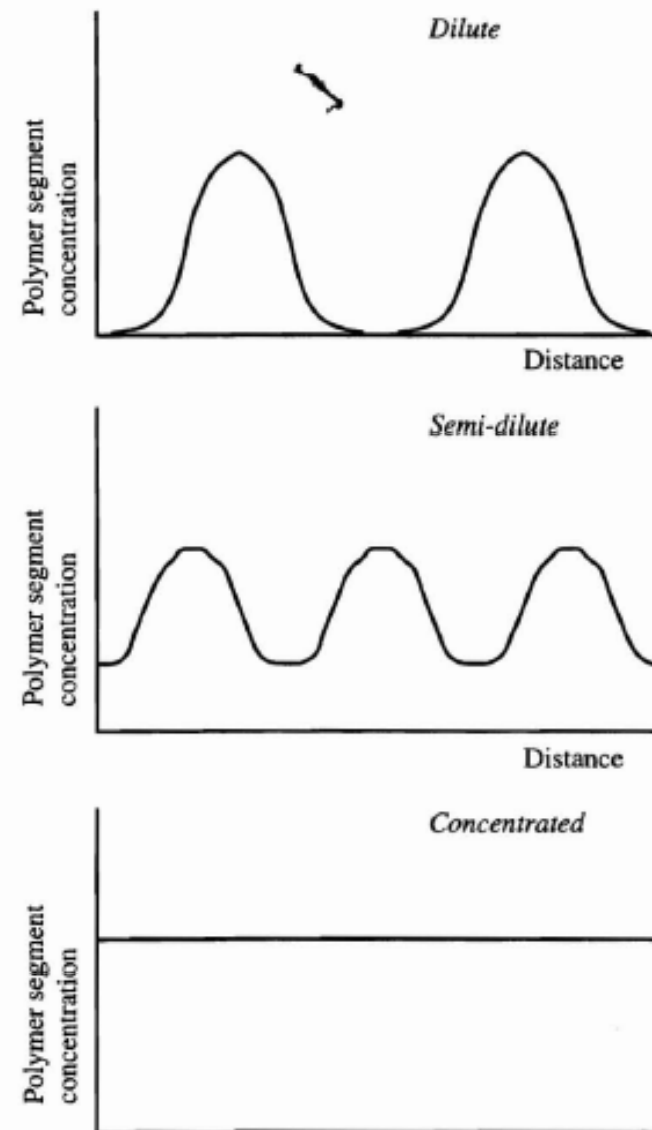
- The total energy:

$$G_{rep} + U_{int} = k_B T v (1 - 2\chi) \frac{N^2}{2r^3} + const;$$

- We expect different behavior according to value  $\chi$ :
  1.  $\chi < 1/2$  polymer chain is expanded (swollen) – good solvent behavior
  2.  $\chi = 1/2$  repulsive effects are cancelled by attractive effects of polymer-solvent interaction. Polymer behaves as ideal random walk: theta condition
  3.  $\chi > 1/2$  attractive effect prevail, polymer forms a globule
- e.g. change in temperature could induce coil-globule transition

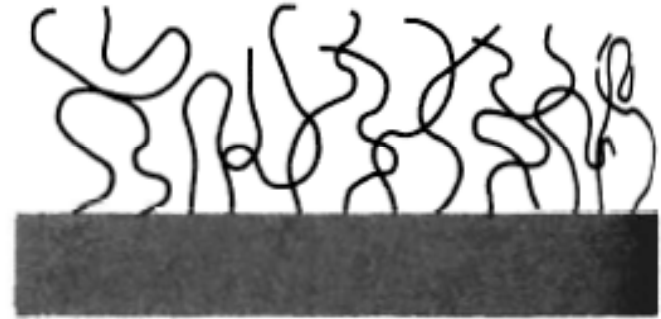
# Flory theorem

- In the polymer melt, chains follow random walk statistics (in other words, upon increasing polymer concentration the effects related to excluded volume disappear).



# Polymers grafted on a surface

- The case of polymer densely grafted on the surface: “polymer brush” with the density of  $\sigma/a^2$  chains per unit area and height  $h$ .



$$\Delta G_{el} = k_B T \frac{r^2}{Na^2} = k_B T \frac{h^2}{Na^2}$$

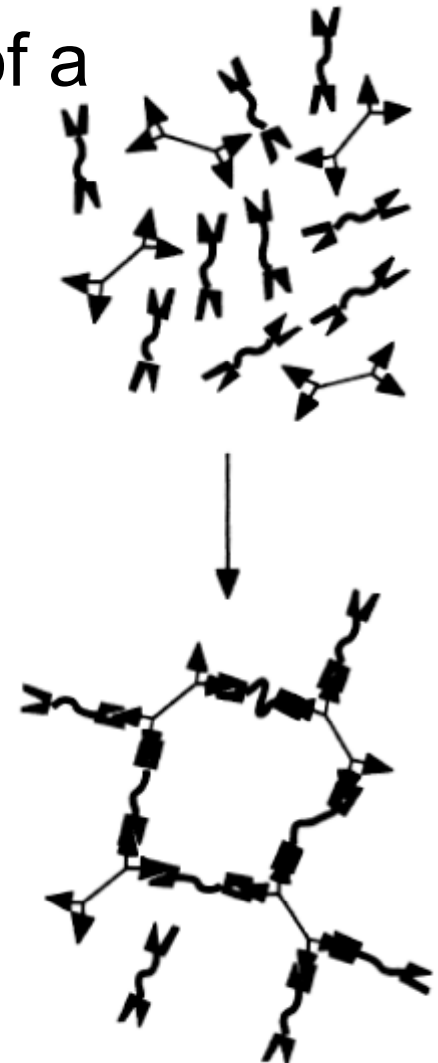
$$\Delta G_{rep} + \Delta U_{int} = k_B T b(1 - 2\chi) \frac{\sigma N^2}{2ha^2}$$

minimizing  
total energy:

$$h \sim [\sigma b(1 - 2\chi)]^{1/3} N$$

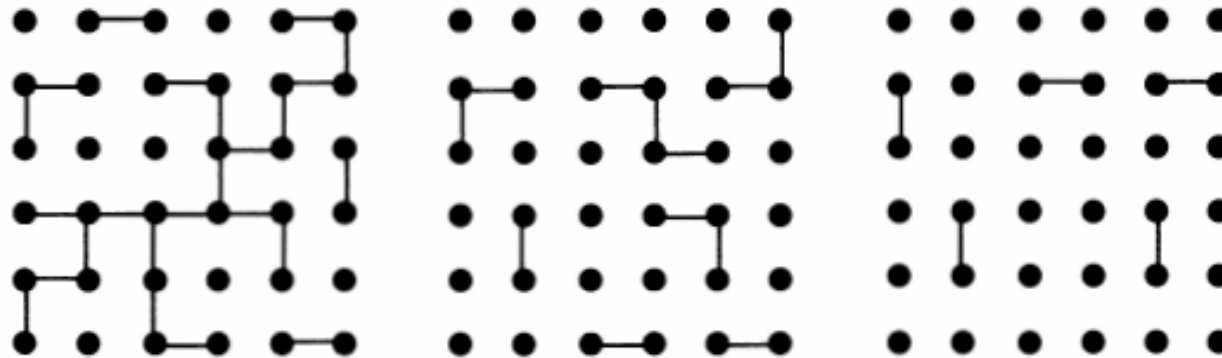
# Gels and gelation

- Gel – material where subunits can bond to each other forming a network
- Gelation or sol-gel transition - formation of a material with non-zero shear modulus
- Gels
  - chemical
    - thermosetting resins (e.g. epoxy)
    - sol-gel glasses (silane polymerisation)
  - physical
    - via microcrystalline regions
    - via microphase separation



# Theory of gelation

- Gelation occurs when an infinitely size cluster is formed



# Gelation theory: Flory-Stockmayer model

- If the probability to form a bond is  $f$  and each point can connect to  $z$  other points, each monomer in  $n$ -th generation is linked to  $f(z-1)$  monomers in  $(n+1)$ th generation. The total number of bonds in  $n$ - generation:

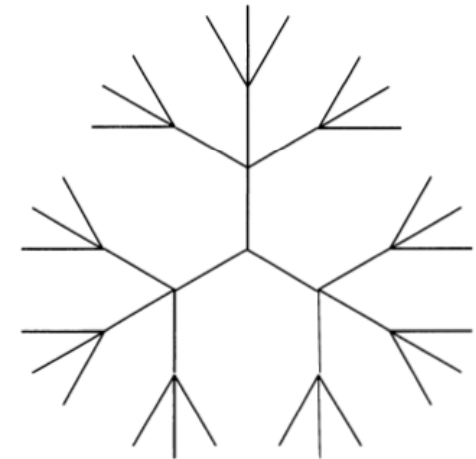
$$N \sim [f(z-1)]^n$$

- So, there is critical probability  $f_c$ -**percolation threshold**:

$$f < f_c \quad N \xrightarrow{n \rightarrow \infty} 0 \quad \text{solution of finite clusters "sol"}$$

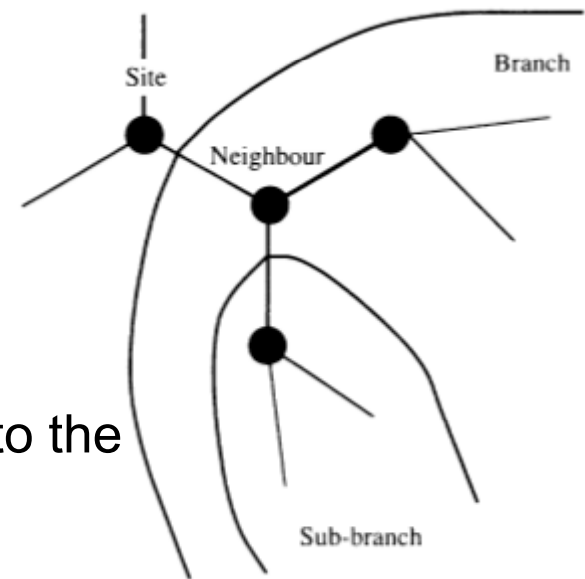
$$f > f_c \quad N \xrightarrow{n \rightarrow \infty} \infty \quad \text{infinite cluster with non-zero shear module}$$

$$f_c = \frac{1}{z-1} \quad \text{near } f_c, \text{ the mean cluster size will diverge}$$



# Gelation theory: Flory-Stockmayer model

- Now, we can calculate the number of bonds inside an infinite cluster (“gel fraction”)
- Let's suppose that the probability for a site to be connected to infinite cluster is  $P$ .  
the probability that the site is not connected via particular branch is  $Q$



- The probability that a neighbour is not connected to the infinite cluster is  $Q^{(z-1)}$

$$Q = (1 - f) + fQ^{z-1}$$

$$Q = 1 \text{ or } Q = \frac{1 - f}{f}$$

- On the other hand,
  - the probability that a site is not connected to infinity is  $Q^z$ .
  - probability to be connected to a particular neighbour but not to infinity  $fQ^z$ .

$$f - P = fQ^z; \quad \frac{P}{f} = 1 - \left( \frac{1 - f}{f} \right)^3 \text{ for } f > f_c \text{ or otherwise } \frac{P}{f} = 0$$

# Gelation theory: Flory-Stockmayer model

- around the gelation point:

$$\frac{P}{f} = 3(f - f_c) + O(f - f_c)^2$$

- generally

$$\frac{P}{f} \sim (f - f_c)^\beta$$

$\beta = 1$  in Flory-Stockmayer theory  
 $\beta = 0.41$  from MC modelling

